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Electrical Conductivities of $SrFeO_{3-\delta}$ Perovskites

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Abstract-A defect chemical model for the behavior of acceptor and donor-doped SrFeO3 as a function of oxygen pressure is proposed. the non-stoichiometric deviation was calculated as a function of oxygen partial pressure, pO_2, at different temperatures. the mathematical approach allows us to calculate the oxygen partial pressure dependent properties of SrFe1-x AlxO3-δ in the range 0.10≤x≤0.30. The results show that the conductivity was dependent of pO_2 and proportional to the dopant concentration. Stability regimes and compensation mechanisms at various oxygen partial pressures and at various temperatures are proposed. This model examines also the charge compensation mechanisms that dominate under the different regimes and their implications for transport properties.

***Index Terms*—Defect modelling; SFA; Nonstoichiometry ; Conductivity.**

I. INTRODUCTION

The search for mixed ionic and electronic conductors (O2_/e_) for solid oxide fuel cell (SOFC) cathodes and oxygen separation membranes has focused strongly on oxides that adopt the ABO3 perovskite-type structure because of their structural flexibility. Not only are these oxides able to accommodate high oxygen deficiencies,1 which yield high rates of oxygen transport, but they can also tolerate significant substitution on both cation sites, whilst still retaining cubic symmetry, which allows their mixed conduction and thermodynamic properties to be tailored.[2][3] Mizusaki and co-workers studied oxygen non-stoichiometry, diffusion and electrical properties of several perovskite-type with the general formula of LaxSr1-xBO3-δ(B=Al, Zr, Bi, Cr, Mn, Fe, Co) by thermo gravimetric method and standard four-probe method.[1][4] Carter et al investigated oxygen transport in perovskite-type and the A- and B- site doping effects on conductivity and other properties[5]. For For these purposes, the condition behavior under law oxygen pressures must be investigated, since if an appreciable electronic conduction arises as a result of defect equilibrium at low pO_2, The electrical conduction is determined by the concentration of present defects in SFA system. several years ago, spinolo et al[1][2] ,suggested a general mathematical method to calculate defect concentrations but without application to actual oxides, A few years later, poulsen[4] has proposed a mathematical approach to calculate the concentration of different species in Ba $Fe_{1-x}Al_x O_{3-\delta}$ system.[4] The purpose of the present work is to establish point defect model equilibrium for $SrFe_{1-x} AL_xO_{3-\delta}$ in the range 0≤x≤0.30 using the non-stoichiometry data that were reported by mizisaki et al[19]. The present defect model will allows us to interpret the thermo-gravimetric results in which oxygen vacancies are assumed for the oxygen deficient condition. The relationship between the obtained resulted and those of conductivity measurements [19][20] will also be discussed The defect model proposed here is considered within the regime that corresponds to oxygen deficiency. Only 1 sub lattice of SrFe1-xALxO3-δ is assumed to be defected. Reduction this system leads to an oxygen deficit in the oxygen sub lattice. Interactions between defects and interstitial oxygen are neglected. This model is a random point defect model, based on the presence of 2 oxidation states of iron ions, Fe+4 and Fe+3, that are populated in various proportions depending on temperature, partial pressure of oxygen and Al- doping. It does not take in to account activity coefficients of all present species. The oxygen non-stoichiometry δ assumes negative values, which is explained by the presence of oxide ion vacancies. In the point defect model, the oxygen vacancies are assumed to be fully ionized at high temperatures. In common with earlier work[22][23], this treatment ignores the Schottky defect equilibrium for which, in any case, the equilibrium constant is not available. For the treatment of point defects in this system we chose the "Kröger-Vink notation"[1][4]. We therefore define vacancies as particles that occupy a defined site in a crystal and that may have a charge. Sites in a crystal are the points where the atoms or the vacancies may be. For a crystal composed of 2 kinds of atoms we have, for example, the "catationic-sites" and the "oxygen-sites". A point, a negative charge, marks the positive excess charge by a dash to distinguish this relative charge from the absolute charge. Adopting this type of notation we obtain on the whole a set of independent equations containing the concentrations of the different species (table 1).

Table 01: Different species used in this defect model with Kroger-vink notation.

Cationic site	anionic sit
$Fe^{\cdot}(Fe^{+4})$	O^{x}_{O}
$Fe^{x}(Fe^{+3})$	VÖ
$Sr'_{Al}(Sr^{+2})$	
$Al'_{Sr}(Al^{+3})$	

The charge-neutrality condition leads to:

$$2[V_{\ddot{O}}]+[Fe^{\cdot}_{Fe}]= [Sr'_{Al}] \quad (1)$$

The value of $[Al'_{Sr}]$ is given by the nominal B site composition x that is

$$[Al'_{Sr}] =x \quad (2)$$

$$[Sr_{Al}] =1 \quad (3)$$

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Because this solid solution is an oxygen deficient type, we have

$$[V_{\ddot{o}}] = \delta \quad (4)$$

$$[O_o^x] = 3 - \delta \quad (5)$$

The non-stoichiometry can be described by the flowing defect reaction

$$2F_{Fe}^{\cdot} + O_o^x \rightleftarrows 2\,Fe_{Fe}^x + V_o^{\cdot\cdot} + 1/2\,O_2\,(g)$$

Oxygen vacancies $V_{\ddot{o}}$ are formed and F^{+4} cations are reduced to F^{+3} at law oxygen partial pressures

$$K_{ox} = \frac{[O_O^{\cdot\cdot}]\,[Fe_{Fe}^{\cdot}]^2}{[V_O^{\cdot\cdot}]\,[Fe_{Fe}^x]^2 P_{O_2}^{1/2}} \quad (6)$$

In order to maintain the fixed A/B ratio, the following equation must maintained:

$$[Fe_{Fe}^x] + [Fe_{Fe}^{\cdot}] + [Al_{Sr}^{'}] = 1 \quad (7)$$

Kröger-Vink notation is used with Sr(+2) Fe(+4)O_3as the reference state.

The experimental non-stoichiometry data as a function of pO_2 are fitted to equation (6) taking the equilibrium constant as a fitting parameter.

From equation (1) (3) (4) (7) we obtain

$$[Fe_{Fe}^{\cdot}] = 1 - 2\delta \quad (8)$$

$$[Fe_{Fe}^x] = 2\delta - x \quad (9)$$

If the set of concentration is accepted, we can insert $(V_o^{\cdot\cdot})$, $(O_o^{\cdot\cdot})$, (Fe_{Fe}^x), and $(Fe_{Fe}^{\cdot})$ into equation (6) and find the oxygen partial pressure that corresponds to the equilibrium concentrations. The calculation is next performed for a new value of $(V_o^{\cdot\cdot})$ until all the concentration interval of interest has been covered.

II. RESULTS AND DISCUSSION

The simulations are normally made for an interval $[V_{\ddot{o}}]$ which corresponds to the experimental data. Solutions are normally generated for a pO_2 range from 10^{-6} to1 atom. The equilibrium constants used in the following are calculated using non-stoichiometric values from TG-data.[4][12][13]

1. Defect concentrations in Sr $Fe_{1-x}Al_x$ $O_{3-\delta}$

For the purpose of comparison, we chose to present for all species and in the Sam defects diagram the concentrations as a function of pO_2. Fig.1 shows the defect diagram for SFA10. At high pO_2, Fe^{+3} ions are oxidized into Fe^{+4} and holes are the dominant defects. The electron concentration decreases while the hole concentration becomes comparable to the point defect concentration. Charge neutrality can only be maintained by decreasing the positively charged metal and increasing the positively charged vacancies.

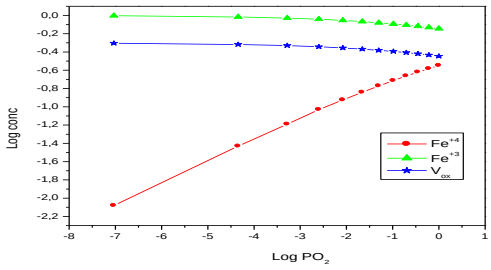

Fig. 1. "Defect diagram for SFA10 at 1173k."

The relative carrier concentration for SFA10 as a function of pressure at different temperatures is shown in Fig. 2. The variation of Fe^{+4} for Sr $Fe_{0.9}$ $Al_{0.1}$ O_3 as a function of partial pressure calculated by Van Hassel et al.[24] is similar to the obtained results (Fig. 1) for SFA10. In the ionic compensation charge, the carrier concentration appears to be in agreement with pO_2 dependence proposed by the model within the temperature range investigated (923K-1223K).

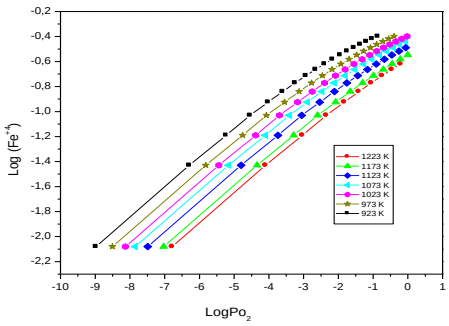

Fig. 2." Relative carrier concentration as a function of pO_2 and temperature for SFA10."

The effect of Al content at 1173 K on the carrier concentration is shown in Fig. 3. The relative carrier concentration appears to have a one-fourth power dependence on pO_2 in the region where oxygen compensation is expected (pO_2>10^{-6}atm). The figure also indicates that the transition from ionic to electronic compensation occurs at relatively higher pO_2 as the Al content increases, as expected from the model. Electron model [3] described very well the non-stoichiometry behavior of SFA system.

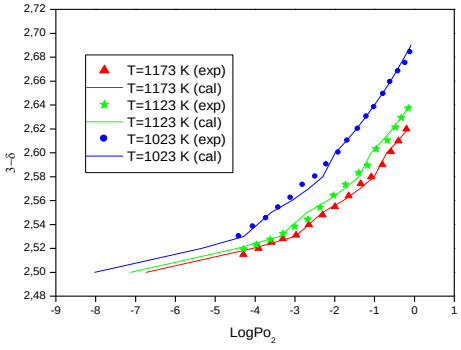

Fig.5. "Simulated oxygen non-stoichiometry of SFA10 as a function of oxygen partial pressure at different temperatures."

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2. Electrical conductivity

Depending on non-stoichiometry, the charge carriers in SFA may be either electrons, holes or both. Therefore, the general expression for the electrical conductivity must involve the components related to both electrons and holes:

$$\sigma = q(n * \mu_n + p * \mu_p) \quad (10)$$

Where q is the elementary charge, μ is the mobility and the subscripts correspond to the specific charge carriers. As seen from equation (1), both the reduced regime and the oxidized regime are governed by the same charge neutrality. Fig. 6 present the dependence of the conductivity on oxygen partial pressure at different temperatures. A good agreement was obtained between the calculated conductivities presented by the solid lines and the experimental data [2][12] at different temperatures. This indicates clearly that the point defect model proposed describes well this material.

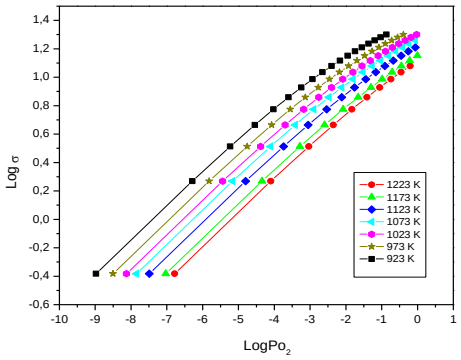

Fig. 6." Electrical conductivity of SrFe0.9Al0.1O3$_{-\delta}$ as a function of po2 at different temperatures."

The decrease of the electrical conductivity with increasing temperature reflects the decrease of the concentration of free electrons, holes and tetravalent Fe ions Fig. 2.the comparison of this behavior with oxygen non-stoichiometry measurements suggests that it is correlated with the increase in the concentration of Fe^{+4}. The effect of temperature on the conductivity for SFA10 is shown in Fig. 6. In the ionic compensation region, the electrical conductivity dependence on oxygen partial pressure proposed by this model is in good agreement with Meadowcroft's Work [21]. This behavior is similar to that observed in Al-doped $SrFeO_3$[25][26] and LSM[17]. As expected, the Figure further indicates that, in the intermediate pO_2 range, where electronic compensation predominates, the conductivity is independent of both temperature and pO_2 .The transition from pO_2 dependence to independence shifts to higher oxygen partial pressure as the temperature of equilibration increases from 923-1223k. These results are in close agreement with the reported conductivity measurements [6][20][21] Since the ionic conductivities of SFA(Fig. 7) are by several orders of magnitude smaller than the electronic conductivities [5][6][12], the experimental determined total conductivities can in good approximation be considered as electronic conductivities. The thermoelectric power was found to be positive in SFA suggesting holes as the dominant charge carriers [5] [16].

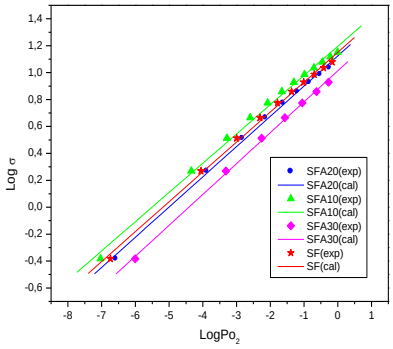

Fig.7." Conductivity isotherms at 1173k for three different compositions SFA10,SFA20 and SFA30."

Faber et al suggested that $SrFe_{0.9}$ $A_{0.1}$ $O_{3-\delta}$ is a p type conductor and its conductivity arises from the presence of multivalent Fe ions due to Al doping [20]. The constant electrical conductivity that exists in the high pO_2 region may be easily understood if it is assumed that acceptors control the carrier concentration and that the electronic compensation predominates. In the lower pO_2 region, oxygen vacancies are formed and the electrical conductivity begins to decrease due to ionic charge compensation. The conductivity mechanism in acceptor-doped strontium iron is due, as mentioned in the literature, to small-polaron hopping irrespective of the kinds of dopants[27][28] .

IV. CONCLUSION

The data obtained from TG experiments support the proposed defect model for the oxidation behavior of Sr $Fe_{1-X}Al_x$ $O_{3-\delta}$. This model indicates that at high pO_2 electronic compensation occurs by a transition of Fe^{+3} to Fe^{+4}, whereas ionic compensation takes place at lower pO_2 by the formation of oxygen vacancies. The oxygen non stoichiometry of B-sile mixed $Sr(Fe,Al)O_3$ strongly depends on the composition of B-site dopants. The stability of the oxide increases for lower Al content at high pO_2. The conductivity of SFA system was mainly due to p-type charge carriers Fe^{+4} which act as traps.

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